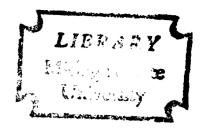
# VAPORIZATION THERMODYNAMICS OF YbBr2

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#### ABSTRACT

#### VAPORIZATION THERMODYNAMICS OF YbBr

By

#### Michael Daniel Gebler

The vaporization thermodynamics of the reaction

$$YbBr_2(\ell) \rightarrow YbBr_2(g)$$

were described over the 1190-1514 K temperature range by the use of a target collection Knudsen effusion technique. The microgram quantities of effusate which plated onto targets were analysed with X-ray fluorescence by the use of an external calibration procedure.

From a plot of the natural logarithm of partial pressures due to YbBr, versus reciprocal temperature the second law enthalpy and entropy at the median temperature were obtained as:  $\Delta H_{1360}^{O} = (66.9 \pm 3.5) \text{ kcal/mole, } \Delta S_{1360}^{O} =$  $(30.0 \pm 2.6)$  eu. Choice of a HgBr<sub>2</sub> model system permitted thermodynamic parameters of YbBr, (g) to be described thus leading to reduction of median temperature values to a reference temperature (298 K).

The estimation of free energy functions by the use of the HgBr, model system and an estimated absolute entropy allowed third law analysis of the vapor pressure data. second and third law values obtained were:

 $\Delta H_{298}^{O}$  (2nd law) = (79.8 ± 4.4) kcal/mole,

 $\Delta S_{298}^{O} = (47.5 \pm 3.9) \text{ eu; } \Delta H_{298}^{O} \text{ (3rd law)} = (72.3 \pm 1.5)$ kcal/mole. The overall consistency of the data was shown by the lack of a trend in third law values of  $\Delta H_{298}^{O}$  over the 324 degree range of the experiment. From the reduced second law values, the enthalpy and entropy and literature values, free energies of formation were calculated as:  $\Delta H_{f298}^{O} \text{ YbBr}_{2}(g) = -(90.2 \pm 0.2) \text{ kcal/mole, } \Delta H_{f298}^{O} \text{ YbBr}_{2}(s) =$ -(170. $_0$  ± 4. $_4$ ) kcal/mole. The second law absolute entropy was determined from the entropy of vaporization and entropy of YbBr, gas (estimated from HgBr, values) as:  $S_{298}^{O}$  YbBr<sub>2</sub>(s) = (29.<sub>0</sub> ± 3.<sub>9</sub>) eu, which combined with literature values of entropy for Yb(s) and  $Br_2(\ell)$  allowed calculation of  $\Delta S_{f298}^{O}$  YbBr<sub>2</sub>(s) = -(21.7 ± 3.9) eu. Values of  $\Delta G_{f298}^{O} \text{ YbBr}_{2}(s) = -(163.5 \pm 4.6) \text{ kcal/mole and } \Delta S_{f298}^{O}$ YbBr<sub>2</sub>(g) =  $(25.8 \pm 3.9)$  eu were determined and subsequently  $\Delta G_{f298}^{O}$  YbBr<sub>2</sub>(g) = -(97.<sub>9</sub> ± 1.<sub>2</sub>) kcal/mole was estimated. By resorting to a PbBr<sub>2</sub> model system to obtain thermodynamic functions for YbBr<sub>2</sub>( $\ell$ ) calculation of  $\Delta H_{V}^{O}$  =  $(58.6 \pm 4.4)$  kcal/mole was made at the normal boiling point of  $(2.0_3 \pm 0.1_1) \times 10^3$  K.  $\Delta S_v^0 = (28._9 \pm 2._2)$  eu was then calculated.

### VAPORIZATION THERMODYNAMICS OF Ybbr<sub>2</sub>

Ву

Michael Daniel Gebler

#### A THESIS

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DEDICATION

To My Parents

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#### TABLE OF CONTENTS

		Pag	e
I.	INT	PRODUCTION	
II.		CORETICAL CONSIDERATIONS NECESSARY TO PORIZATION STUDIES	
	Α.	Modes of Vaporization and Phase Rule Considerations	
		1. Modes of Vaporization 3	
		2. Phase Rule	
	в.	Methods of Vapor Pressure Measurement 4	
	c.	Vapor Pressure Determinations by the Knudsen Effusion Method; the Use of Target Collection Procedures 5	
	D.	Assumptions of the Knudsen Method 9	
	Ε.	Limitations of the Knudsen Effusion Method	
		1. Non-Ideal Cells 10	
		a. Clausing Correction Factor 10	
		b. Correction for Orifice Effects by	
		Proper Choice of Collection Geometry	
		2. Vaporization Coefficient 12	
		3. Striking Coefficient	
		4. Interactions with Knudsen Cells 13	
		5. Non-Ideal Gas	

#### TABLE OF CONTENTS (Cont.)

		Pag	је
	F.	Analysis of Targets; X-Ray Fluorescence 15	;
		1. X-Ray Fluorescence: Basic Method 16	;
	G.	Temperature Measurement	;
	н.	Thermodynamic Calculations 19	)
		1. Second-Law Calculations 19	)
		2. Third Law Calculations 21	_
III.	EXP	ERIMENTAL	}
	A.	Preparation of YbBr <sub>2</sub>	3
	В.	Chemical Analysis	ļ
		1. For Ytterbium	ļ
		2. For Bromide	ŀ
	c.	X-Ray Powder Diffraction Analysis 25	;
	D.	Target Collection Apparatus 25	;
	E.	Effusion Cells	}
	F.	Temperature Measurement	}
	G.	Heat Source	<b>)</b>
	н.	Targets	)
	I.	Orifice Measurement	)
	J.	Miscellaneous Measurements and Equipment . 30	)
	к.	Procedure for Vaporization	)
	L.	X-Ray Fluorescence Analysis of Mass of Effusate	Ĺ
		1. X-Ray Fluorescence Unit 31	L

#### TABLE OF CONTENTS (Cont.)

		Pag	е
		2. Calibration of X-Ray Fluorescence Unit	
		3. Target Analysis	,
IV.	RES	SULTS AND DATA REDUCTION	r
	A.	Elemental Analysis of YbBr <sub>2</sub> 35	
	В.	X-Ray Fluorescence Calibration 35	1
	c.	Thermocouple Calibration	
	D.	Vaporization Experiments	
	Ε.	Mode of Vaporization	;
	F.	Vapor Pressure Equation $\Delta H^O$ and $\Delta S^O$ at Mean Temperature of Vaporization Experiments	
	G.	Estimation of Thermodynamic Values 42	ı.
	н.	Second Law Data Reduction to 298 K 44	:
	ı.	Value of Absolute Entropy 45	,
	J.	Estimation of fef and Afef for YbBr <sub>2</sub> 45	
	к.	Data Reduction by the Third Law Procedure. 46	į
	L.	Other Thermodynamic Parameters for YbBr <sub>2</sub> . 47	
		1. From Literature Estimates and Measured Data 47	
		2. From Extrapolation of the Vapor Pressure Equation 48	,
	M.	Note on Errors 49	I
v.	DIS	SCUSSION AND SUGGESTIONS 50	ı
	Α.	Preparation of YbBr <sub>2</sub> 50	J

#### TABLE OF CONTENTS (Cont.)

		Page
В.	X-Ray Fluorescence Procedures	50
c.	Target Collection Knudsen Effusion Technique	52
D.	The Use of Thermodynamic Approximation	5 <b>4</b>
	1. Absolute Entropy Approximation	54
	2. YbBr <sub>2</sub> Data	54
Ε.	Suggestions for Future Research	58
וקקם	ERENCES	61

#### LIST OF TABLES

		Page
I.	Analytical Results	35
II.	X-Ray Fluorescence Calibration Results	36
III.	Vaporization Experiments	38
IV.	Data Summary for Vaporization Thermodynamics of Eu(II) and Yb(II) Halides	57

#### LIST OF FIGURES

		Page
1.	Effusion cell-target collection geometry	. 8
2.	X-ray fluorescence spectrometer	. 17
3.	Symmetric effusion cell	18
4.	High vacuum Knudsen effusion apparatus	. 26
5.	Effusion cell-heating oven arrangement	. 27
6.	Pressure of YbBr <sub>2</sub> (g) in equilibrium with YbBr <sub>2</sub> (s)	. 39

#### LIST OF APPENDICES

		Page
Α.	Collected Vaporization Data and Third Law Enthalpies for YbBr <sub>2</sub>	65
В.	Enthalpy, Entropy, Free Energy Functions of $YbBr_2(s,\ell)$	67
c.	Free Energy Function Changes for the Vaporization of $YbBr_2(\ell)$	68
D.	Thermodynamic Functions	69
Ε.	X-Ray Powder Diffraction Patterns	70

#### CHAPTER I

#### INTRODUCTION

Experimentally determined values for such thermodynamic parameters as enthalpies of fusion, vaporization and formation, as well as for heat capacities are not available for many of the rare earth halides. Indeed, as late as 1964 Novikov and Polychenok<sup>1</sup> observed the lack of experimental conformation for many values found in tables of estimates such as those by Brewer et al.<sup>2,3</sup> especially for divalent compounds. Since then much interest has centered around the trivalent state. Vapor pressure values and sublimation thermodynamics for most of the lanthanide(III) fluorides have been established.<sup>4-10</sup> Some vapor pressure values are known for lanthanide trichlorides and tribromides.<sup>11-15</sup>

For lanthanide halides in the less common divalent state Polychenok and Novikov<sup>16</sup> have reported thermodynamic values for SmCl<sub>2</sub>, EuCl<sub>2</sub>, and YbCl<sub>2</sub> but they assert that the "boiling point" method they used tends to be inherently inaccurate. Haschke and Eick<sup>17</sup> and Hariharan<sup>18</sup> have used the Knudsen effusion method to determine the vaporization thermodynamics of EuCl<sub>2</sub>, EuI<sub>2</sub>, and EuBr<sub>2</sub> and Hariharan,

Fishel, and  $\operatorname{Eick}^{19}$  used the same technique to describe the YbCl<sub>2</sub> system.

It was the intention of the present work to establish experimental values of the enthalpy of vaporization and entropy of vaporization of YbBr<sub>2</sub> by the use of procedures similar to those applied by Haschke, Eick, and Hariharan to the analogous EuBr<sub>2</sub> system.

#### CHAPTER II

## THEORETICAL CONSIDERATIONS NECESSARY TO VAPORIZATION STUDIES

#### A. Modes of Vaporization and Phase Rule Considerations

#### 1. Modes of Vaporization

When heated to a specific temperature a substance will vaporize in either a congruent or incongruent manner. A substance which vaporizes congruently yields a vapor that has the same chemical composition as that of the condensed phase from which it was obtained; incongruent vaporization results when a condensed phase gives rise to a vapor of different composition from that of the condensed phase.

#### 2. Phase Rule

The phase rule:

$$F = C - P + 2 \tag{II-1}$$

establishes many useful relationships for vaporization studies. In II-1 the number of degrees of freedom, F,

is related to the number of components in a system, C, and the number of phases, P.

For a binary system of a condensed (either liquid or solid) phase and a congruently derived vapor, P will equal two in II-1 above. The number of components, C, will equal one since the vapor and condensed phase are chemically equivalent so that the number of degrees of freedom for the system will be one. Therefore, to obtain a parameter such as vapor pressure it is only necessary to fix one experimental condition such as temperature.

#### B. Methods of Vapor Pressure Measurement

A number of methods are available to establish the vapor pressure of a system. 20 These fall into two classifications, the absolute, which include the static and boiling point methods, and the non-absolute which are the effusion and transpiration techniques. The non-absolute procedures rely on the kinetic theory of gases which necessitates the assumption of a molecular weight of the vapors, hence they are not applicable in those systems where vaporization of fragmented or polymeric components invalidates the assumption. However, for less volatile or chemically reactive substances they are the usual methods of choice because they allow the vaporizing system to operate at lower overall temperature than the absolute methods and because they operate at high vacuum.

The effusion techniques include an open, the Langmuir free evaporation, and a closed method, the Knudsen effusion procedure. Both are theoretically and in essence the same. The former relies on vaporization from a surface of known area and the latter on the amount of effusate which can be lost from an otherwise closed system without significantly shifting the equilibrium which is established between the condensed phase and the vapor. Both have advantages and disadvantages, but since the surface area of a vaporizing substance is difficult to determine accurately the latter method, the Knudsen Effusion technique, was chosen for our vapor pressure determinations.

# C. Vapor Pressure Determinations by the Knudsen Effusion Method; the Use of Target Collection Procedures

When certain conditions (see below) are met the kinetic theory of gases allows establishment of vapor pressures according to a method first put forth by Knudsen. The Knudsen Effusion technique requires that a small "ideal" (see below) orifice be placed in a sample cell which contains a condensed phase and its vapor in equilibrium such that the amount of effusate which escapes from the orifice does not appreciably shift the equilibrium. The number of molecules striking a unit area

of the interior of the cell per unit time, Z, is proportional to the number of molecules per unit volume, n, and the average molecular velocity,  $\bar{v}$ . Precisely:

$$Z = n\overline{v}/4 \text{ molecules cm}^{-2} \text{sec}^{-1}$$
 (II-2)

If an "ideal" orifice, that is, a small, infinitesmally thin, circular orifice is placed in the container so that a small portion of the vapor can escape into a perfect void (guaranteed by high vacuum) in such a way as not to affect equilibrium within the cell, then the number of molecules escaping through the orifice of area A per second will be given by:

$$N_{O} = A_{O}Z \tag{II-3}$$

Now if a circular target is placed at a distance d above the orifice so that the center of the target is coaxial with the center of the orifice, and if the target has radius r, then the fraction of molecules striking the target can be determined since the effusing vapor follows the cosine distribution law:

$$dN = \pi^{-1}N_{o} \cos \theta d\omega \qquad (II-4)$$

Here  $N_O$  is the total flux at the orifice,  $\theta$  is the angle between the perpendicular and the axis of  $d\omega$  (see Figure 1), the solid angle formed by the effusate which is subtended by a target of area dN. Upon substitution in terms of r, d,  $A_O$ , and Z into the cosine distribution law and integration of the value over the space above the orifice one obtains:

$$N = ZA_o(r^2/(d^2+r^2)) \text{ molecules sec}^{-1}$$
 (II-5)

and as long as pressure remains below 10<sup>-3</sup> atm in the sample cell to meet Knudsen conditions the gas can be assumed ideal so that the ideal gas law can be applied.

When  $\bar{\bf v}=\left(8{\rm RT/\pi M}\right)^{1/2}$  is substituted into II-2 and the Z so obtained used in II-5 above, multiplication of II-5 by the total time effusate is allowed to strike the container, t, allows the inclusion of the ideal gas law assumption to yield the following equation for equilibrium vapor pressure.

$$P = [W/A_0t] [2\pi RT/M]^{1/2} [(d^2+r^2)/r^2]$$
 (II-6)

If W, the mass in grams of effusate of molecular weight M, is collected in t minutes on a circular target of radius r cm placed d cm above an orifice of  $A_O$  cm<sup>2</sup> area, and R is

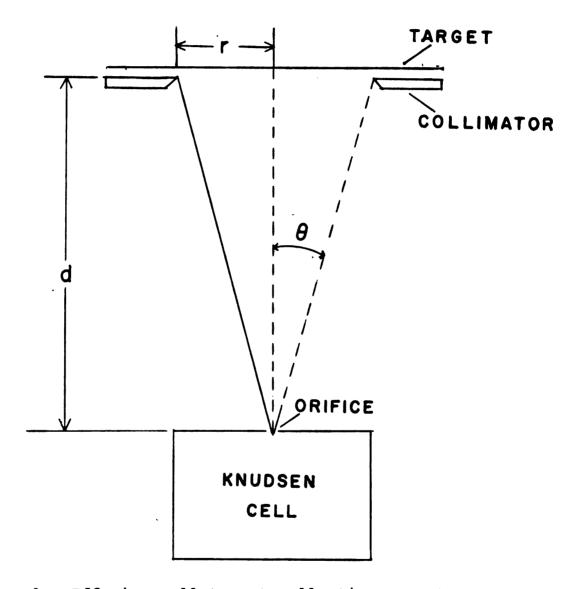


Figure 1. Effusion cell-target collection geometry.

defined in ergs deg<sup>-1</sup> mole<sup>-1</sup>, then P is obtained in units of dynes cm<sup>-2</sup>, a unit which is easily converted into atmospheres. Substitution of appropriate constants into II-6 yields:

$$P_{atm} = [3.76 \times 10^{-4} \text{ W/A}_{O}t] [T/M]^{1/2} [(d^2+r^2)/r^2]$$
 (II-7)

If a molecular weight of the effusate is known or can be assumed, and if the weight of effusate plated on a target can be determined, then measurement of time in minutes, area of the orifice in cm<sup>2</sup>, and temperature in degrees Kelvin will yield the vapor pressure of the system when the geometry of the system (hence d and r) is known.

#### D. Assumptions of the Knudsen Method

Aside from the fundamental assumption that the orifice is "ideal" the system for Knudsen effusion relies on other assumptions many of which are required by the kinetic theory of gases. These include:

- 1. Isothermal conditions exist about and within the cell.
  - 2. Molecules are point masses.
  - 3. Isotropy of gas exists in the cell.
  - 4. Molecular velocity distribution is Maxwellian.
- 5. There are no molecular interactions in the gas phase.

- 6. Orifice walls do not return any molecules to the cell.
- 7. Molecules do not return to the cell once through the orifice.
  - 8. Loss of effusate occurs by vapor transport only.
  - 9. Equilibrium pressure is maintained within the cell.
  - 10. No molecular collisions occur in the orifice.

#### E. Limitations of the Knudsen Effusion Method

All of the assumptions stated above would be valid in an ideal system. However, in a real system some of the assumptions can only be approximated and the resulting error must be corrected for by changing equation II-7 so that it includes a number of correction factors.

#### 1. Non-Ideal Cells

#### a. Clausing Correction Factor

The basic assumption that the orifice is ideal, which implies infinitesmally thin, is violated immediately in a real system since it is impossible to obtain such an opening in an absolute sense. Therefore, a certain resistance to molecular flow occurs, the so-called channeling effect. Clausing demonstrated 23 that the effect is a

function of orifice geometry and that one can apply a correction factor (the Clausing Factor),  $W_{\rm O}$ , to the usual vapor pressure equations, II-6, II-7. He obtained these correction factors in terms of orifice radius and orifice length. Inclusion of  $W_{\rm O}$  in II-7 makes the vapor pressure equation:

$$P_{atm} = [3.76 \times 10^{-4} \text{ W/A}_{o}t] [T/M]^{1/2} [(d^2+r^2)/r^2] [1/W_{o}]$$
(II-8)

# b. Correction for Orifice Effects by Proper Choice of Collection Geometry

The limits of the cosine distribution law for real knife-edged conical orifices have been examined by Ward.  $^{24}$  He demonstrates through experiment and calculation that when a target geometry is chosen such that only small angles of  $\theta$  are subtended there is little or no effect on the cosine distribution law for conical orifices. Therefore, by simply choosing the target geometry such that  $\theta$  remains small and by using a conical orifice one need not apply a correction factor. Indeed, if the target is small enough or at great enough distance to receive about 1% or less  $^{24}$  of the effusate the Clausing correction factor can be taken as unity and vapor pressures will be given directly by II-7.

#### 2. Vaporization Coefficient

Another of the Knudsen assumptions requires that the vaporizing system be at equilibrium. That is, the vaporization coefficient  $\alpha_{_{\rm V}}$ , defined as the rate of vaporization compared to the equilibrium rate, is equal to the condensation coefficient  $\alpha_{_{\rm C}}$ , defined as the fraction of vapor molecules which recondense. Since the Knudsen effusion procedure allows a certain portion of vapor to escape a steady state loss might arise. The pressure given by II-7,  $P_{_{\rm m}}$ , as measured will not be equal to the true equilibrium vapor pressure  $P_{_{\rm C}}$ . Motzfeldt  $P_{_{\rm m}}$  to  $P_{_{\rm C}}$  assuming that  $P_{_{\rm C}}$  and that a resistance similar to a Clausing orifice factor develops along the cell wall. In its mathematical form the equation is:

$$P_e = P_m [1 + f(\frac{1}{\alpha} + \frac{1}{W_a} - 2)]$$
 (II-9)

Here f =  $W_O A_O/A$  with  $A_O$  being the orifice area and A the sample surface area,  $W_O$  is the Clausing orifice factor,  $W_A$  is a "Clausing factor" of the cell body and  $\alpha = \alpha_V = \alpha_C$ . From equation II-9 it is evident that when  $A_O/A \leq .01$ ,  $P_M \approx P_C$  provided  $\alpha$ , the vaporization coefficient does not differ significantly from unity. Indeed, early work by Rosenblatt<sup>26</sup> has established that finely divided samples

(or those with large surface areas) usually have vaporization coefficients which approach unity. However, it appears that the vaporization coefficient need not strictly approach unity with increasing sample size but is dependent on a number of experimental and system factors which include surface effects of the samples. An extensive review of the vaporization coefficient problem was given by Work. 27

#### 3. Striking Coefficient

If all the effusate striking a target does not adhere to it another source of error arises. One can determine the amount of material which does strike to the target by placing a chilled disc with a hole in its center in front of the target. Any molecules not striking to the target will be reflected back by the cosine law and a fraction of these will adhere to the disc. By measurement of the amount on the target and the amount on the disc a correction factor can be obtained.

#### 4. Interactions with Knudsen Cells

Steps must be taken to insure that the vaporizing compounds do not interact with the cell material. Simple mass difference determinations of an empty cell before

vaporization and after 100% vaporization of the cell's content will lend more qualitative insight but the method is limited by the sensitivity of the balance used for mass determination.

Usually it is sufficient to rely on X-ray powder diffraction patterns (hereafter called "patterns") of the material remaining in the cell after the vaporization. comparing the patterns of the residue after vaporization to those of the starting material two determinations can be made. If the patterns before and after vaporization are the same then, reasonably, one can assume no interaction between the cell and the sample. Further, the vaporization was probably congruent. If they are not the same then chemical analysis (using wet chemical and/or X-ray fluorescence methods) is necessary to establish what materials are present in the residue. If analysis shows that the residue is simply the starting material's elements in different mole fractions then incongruent vaporization should be suspected, but if the residual material contains a compound composed in part of the cell's elements then one must try to describe the chemical process taking place to ascertain a correction factor. If a correlation cannot be found, then measured values of vapor pressure cannot be related to a specific reaction.

#### 5. Non-Ideal Gas

Since the kinetic theory of gases is fundamental to the Knudsen effusion theory most of the assumptions mentioned in Section D are required to insure that its principles are not violated. Many of these assumptions center around pressure build-ups in the cell. If free molecular flow is to be maintained the pressure must not reach the point that molecular interactions occur and a "viscous flow" of molecules results in the molecular flux. Experiments conducted by Meyer  $^{28}$  show that for orifices of areas between  $10^{-4}$  and  $10^{-5}$  cm $^2$  the pressure should not exceed  $5 \times 10^{-3}$  atm if free molecular flow is to be maintained.

#### F. Analysis of Targets; X-Ray Fluorescence

One must ascertain the amount (mass) of material plated on the targets during the vaporization process before II-7 can be used to compute vapor pressures. Since the amount of plated material is usually maintained in the microgram region to avoid adhesion problems, X-ray fluorescence is uniquely suited to the task of mass determination.

#### 1. X-Ray Fluorescence: Basic Method

When white X-radiation is allowed to shine on a sample it excites inner shell electrons to higher energy levels. Those electrons which return to the ground state emit photons of characteristic wavelength -- the basic fluorescence. The radiation thus given off can be analysed by allowing it to diffract from a suitable crystal and by arranging a detector (such as a scintillation counter) such that it picks up the particular wavelength given off by the element being analysed for. The fluorescence X-radiation is analysed according to the Bragg equation:

$$n\lambda = 2d\sin \theta$$
 (II-9)

where n is the order of diffraction (usually n=1),  $\lambda$  is the wavelength of the characteristic radiation, d is the inner planar spacing of the analysing crystal and  $\theta$  is the angle of incidence. Placement of the detector at  $2\theta$  allows the correct geometry for analysis (see Figure 2).

#### G. Temperature Measurement

Figure 3 shows a symmetric effusion cell. When the top cavity is used as a sample container and the entire

# Spectrometer X-ray Fluorescence

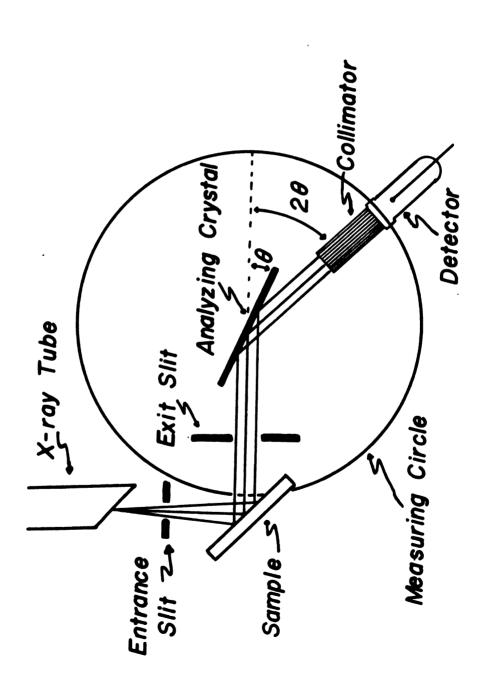
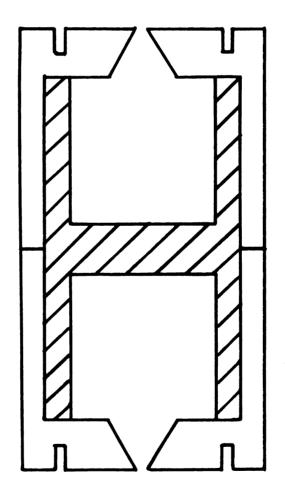


Figure 2. X-ray fluorescence spectrometer.

# SYMMETRICAL EFFUSION CELL WITH SAMPLE AND OPTICAL CAVITIES



6/68

Figure 3. Symmetric effusion cell.

cell maintained in an isothermal environment the bottom cavity should display a temperature identical to that of the top cavity. By placing a thermocouple in the bottom cavity the temperature of the vaporization process can be determined directly.

#### H. Thermodynamic Calculations

#### 1. Second-Law Calculations

With only one degree of freedom available to a vaporizing system a free energy relationship would be expected between temperature and vapor pressure. That is since:

$$\Delta G_{T}^{O} = -RT \ell n P_{T}$$
 (II-10)

and

$$\Delta G_{\mathbf{T}}^{\mathbf{O}} = \Delta H_{\mathbf{T}}^{\mathbf{O}} - \mathbf{T} \Delta S_{\mathbf{T}}^{\mathbf{O}}$$
 (II-11)

we have

$$-\ln P_{T} = (\Delta H_{T}^{O}/RT) - \Delta S_{T}^{O}/R \qquad (II-12)$$

where  $P_T$  is the equilibrium vapor pressure given by II-7 above. From II-12 it can be seen that a plot of  $\ell n P_T$  versus 1/T by least squares regression will yield the relationship of  $\ell n P_T$  = m/T = b so that enthalpies are given by the slope of the line:

$$\Delta H_{T}^{O} = -Rm \qquad (II-13)$$

and entropies are given by the y-intercept:

$$\Delta S_{T}^{O} = Rb$$
 (II-14)

The entropy and enthalpy values thus obtained are usually considered those of the mean temperature of the study.  $^{29}$ 

These values are reduced to a reference temperature according to the following:

$$\Delta H_{298}^{O} = \Delta H_{T}^{O} + \int_{T}^{298} CpdT$$
 (II-15)

and

$$\Delta S_{298}^{O} = \Delta S_{T}^{O} + \int_{T}^{298} Cp/T dT$$
 (II-16)

where it is usually necessary to express the heat capacities in their analytical forms:

$$C_{p} = a + bT$$
 (II-17)  
 $C_{p} = a + bT + cT^{-2}$ 

#### 2. Third Law Calculations

The third law treatment utilizes a free energy function, fef, to reduce thermodynamic data to a reference temperature which results in a value of  $\Delta H_{298}^{O}$  for each data point. The fef can be defined as:

$$fef = (G_T^O - H_{298}^O)/T$$
 (II-18)

or

$$fef = (H_T^O - H_{298}^O)/T - S_T^O$$
 (II-19)

Afef values are calculated from those of fef for each product and reactant by:

$$\Delta fef = \sum_{i} v_{i} fef_{i} - \sum_{j} v_{j} fef_{j}$$
 (II-20)

1

where i refers to the products and j refers to the reactants. Once determined,  $\Delta$  fef values lead to  $\Delta H_{298}^{O}$  values from:

$$\Delta fef = (\Delta G_{T}^{O} - \Delta H_{298}^{O})/T \qquad (II-21)$$

or through

$$\Delta H_{298}^{O} = -(\Delta fef + R \ell n P_{T}) T \qquad (II-22)$$

when II-10 is substituted into II-21. The advantage of the use of third law treatment is that, although it does not give values of  $\Delta S_{298}^{O}$  as the second law does, the treatment results in a value of  $\Delta H_{298}^{O}$  for each data point. Analysis of the values thus obtained allows any trend in  $\Delta H_{298}^{O}$  to be displayed. A trend might arise from systematic error in either pressure or temperature measurement or in the computation of free energy function changes. A large difference in second-law and third law values indicates error in measurement of the parameters of vaporization or indeed, in the basis definition of the vaporization process itself.

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#### CHAPTER III

#### EXPERIMENTAL

# A. Preparation of YbBr $_2$

YbBr, was produced by following the general procedure for rare earth dichloride preparation put forth by DeKock and Radtke. 30 The sesquioxide of ytterbium (99.99% Research Chemicals, P.O. Box 14588, Phoenix, Arizona) was dissolved in approximately 150-200 ml of 4.5  $\underline{N}$  HBr along with  $\mathrm{NH}_{\Delta}\mathrm{Br}$  (Matheson Coleman and Bell) and metallic zinc (Baker). One to three grams of the oxide was used with sufficient quantities of the other reagents to give a ratio of twelve moles to ammonium salt and one and two tenths mole of zinc bromide per mole of rare earth tribromide formed in the first step of the reaction. solution was evaporated to dryness and the dried material transferred to a carbon boat which contained an excess of zinc metal as a reducing agent. After drying under vacuum and low (200°C) temperature, the material was melted in an inert atmosphere to effect reduction. The excess NH, Br was sublimed off and a second attempt was made to remelt the remaining contents of the boat by heating to 500-600°C to insure all of the tribromide was reduced to the dibromide. Next the Zn and ZnBr<sub>2</sub> were vaporized under high vacuum. The greenish-yellow crude YbBr<sub>2</sub> remaining was transferred to an outgassed molydenum crucible and heated by induction in a high vacuum so that it distilled onto a high vacuum so that it distilled onto a quartz condenser (distillation temperature approximately 1145°C).

#### B. Chemical Analysis

#### 1. For Ytterbium

A 50-80 mg sample of distilled  $YbBr_2$  was placed in a crucible and fired directly to the oxide by heating to  $1000^{\circ}C$  in a muffle oven (Thermolyne Model F-Al620) for 2-3 hours.

#### 2. For Bromide

A 60-80 mg sample of YbBr $_2$  was dissolved in dilute nitric acid solution. A 0.1 N solution of silver nitrate was stirred into precipitate Br $^-$  as AgBr. The solution was heated to boiling for 1-2 minutes and allowed to settle overnight. A few drops of AgNO $_3$  were added to the solution above the precipitate to insure complete precipitation

before the solution was filtered into a sintered glass crucible. The material so collected was dried at 110°C for 1½ hours and then weighed as the bromide of silver.

## C. X-Ray Powder Diffraction Analysis

Samples of YbBr $_2$  and the residue left in sample cells after vaporization were prepared for powder X-ray diffraction analysis by sealing small amounts of each substance in plastic bags in a dry box. A Haegg Type Guiner forward focusing camera of 80 mm radius and a Ca K $_{\alpha}$ l radiation source ( $\lambda_{\alpha 1}$  = 1.54051 Å, t = 24±1°C) powered by a Picker 80913 generator was used to obtain the powder patterns.

### D. Target Collection Apparatus

Kent<sup>31</sup> has described the general setup used during the work. Basicly it is a glass vacuum line (Figure 4) in which the effusion cell is supported above a boron nitride table by tungsten rods. A similar set of rods support a molydenum oven arranged symmetrically about the cell (Figure 5). Directly above the oven-cell arrangement the glass line supports a target magazine fitted with a liquid nitrogen dewar so that the targets can be cooled. The line has apparati which allow target changes while the

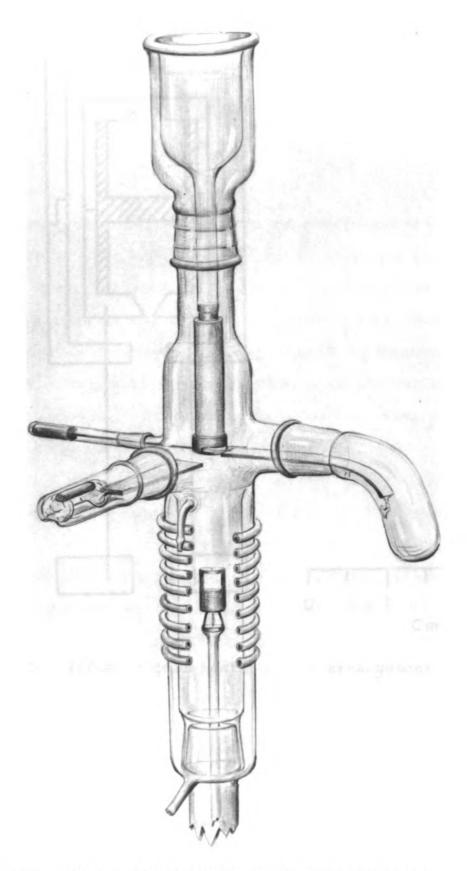


Figure 4. High vacuum Knudsen effusion apparatus.

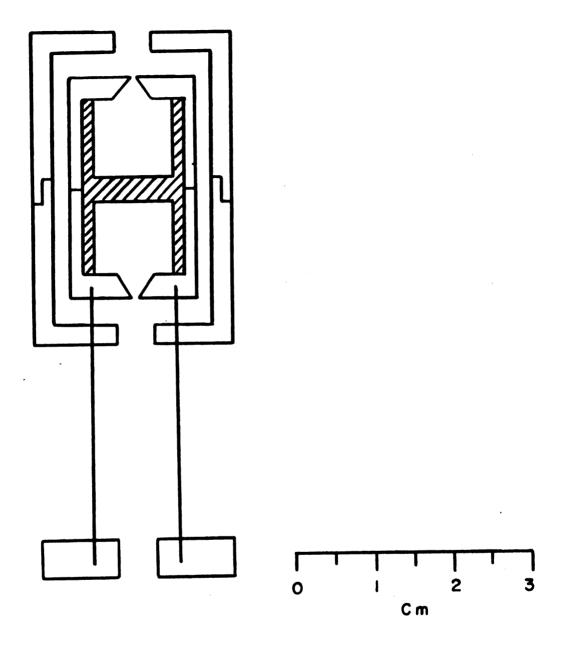


Figure 5. Effusion cell-heating oven arrangement.

system is closed by the use of magnets and iron mechanical parts built into its side. A mercury diffusion pump provides the high vacuum necessary for the experiments.

#### E. Effusion Cells

In every vaporization experiment except one the effusion cells were made of molydenum; the basic design of which was given in Figure 3. As stated before, their symmetric nature allows the bottom cavity to be used as a chamber into which the temperature sensing thermocouple is placed while vaporization occurs from an identical top cavity. In one case a carbon insert was used with molydenum end caps in the usual cell arrangement.

#### F. Temperature Measurement

A two foot thermocouple supplied by Omega Engineering was calibrated against the melting point of National Bureau of Standards copper, and the boiling point and ice point of water. The thermocouple was then used in conjunction with a Numetron 914 Numeric Display digital potentiometer (Leeds and Northrup) to read temperatures directly in degrees centigrade.

#### G. Heat Source

The oven cell arrangement was heated to and held at constant temperature by a 20-kva Thermonic induction generator coil which was situated as symmetrically as possible about the cell.

#### H. Targets

Targets for the vapor study were made of an aluminum backing 2.7 cm in diameter and 0.46 cm thick with a 2.08×0.19 cm circular recession machined in one side. A thin platinum disc was supported in the recession by a stainless steel retaining ring. The platinum discs were cleaned prior to vaporization runs by degreasing with petroleum ether, washing with detergent, scouring with steel wool, and boiling in dilute nitric acid. The aluminum backings were treated in a similar manner except they were boiled in distilled water. The retaining clips were scoured with detergent and steel wool. All parts were rinsed in distilled water and dried at 130°C.

#### I. Orifice Measurement

The area of the orifices used in the experiments was measured precisely by finding the area of an enlarged photomicrograph (Bausch and Lamb Dynazoom Metallograph fitted with Polaroid attachment) with a polar planimeter (Keffel and Esser) and then multiplying by an enlargement factor of  $0.25 \times 10^{-4}$ . The enlargement factor had been found by calibrating the Metallograph's internal scale with a micrometer slide (American Optical Co.).

#### J. Miscellaneous Measurements and Equipment

Time was measured to ±0.01 minutes by a Lab Con timer.

A precision cathetometer with a readibility of 0.005

cm (Gaertner Scientific) was used to measure the cell to target distance.

## K. Procedure for Vaporization

Approximately 0.25 g of YbBr<sub>2</sub> was placed in the sample cavity of an outgassed effusion cell in an argon atmosphere glove box. A drop of dried fluorolube oil was placed over the orifice and the cell quickly transferred to the vacuum line. The cathetometer was used to obtain the height of

the top of the cell before the remainder of the glass vacuum system was assembled. After the apparatus had pumped down to  $<10^{-4}$  torr a reading was made of the height of the flat spot on the target magazine. The assembly was heated to  $200-300\,^{\circ}\text{C}$  to insure outgassing before the actual vaporization experiment began. When the vacuum had reached  $<10^{-5}$  torr liquid nitrogen was added to the target magazine's trap and the system heated to vaporization run temperatures. Targets were exposed for periods of time sufficient to plate between two and twelve micrograms (as ytterbium) on them (times were found by trial and error).

## L. X-Ray Fluorescence Analysis of Mass of Effusate

## 1. X-Ray Fluorescence Unit

``

The targets were analysed by a 4-position Norelco Universal Vacuum spectrograph with a broad focus tungsten X-ray tube powered by a Norelco XRG 5000 X-ray generator. The spectrograph was set at a 20 value of 28.88°, the optimum setting for the  $L_{\alpha 1}$  radiation of Yb for our particular instrument.

#### 2. Calibration of X-Ray Fluorescence Unit

Since the targets collect between 2 and 12 micrograms of effusate it is necessary to know how many counts per microgram of Yb the detector will "see". To find the value each of six targets was counted blank to determine its background fluorescence at the particular settings used in the experiments. Next each target was plated with 49.6  $\lambda$  of solution delivered by a precision micropipet (Misco). The solutions thus deposited contained between 2.7 and 12.11 micrograms of Yb made up as a six member series from a stock solution obtained by dissolving a carefully weighed sample of Yb<sub>2</sub>O<sub>3</sub> (99.99% Research Chemicals) in HCl, boiling the majority of acid off, and then diluting to 500 ml. The solutions deposited on the target were evaporated to dryness under a high intensity light. Much skill was necessary to perfect the art of plating so that consistent values were obtained target to target, but the remainder of the procedure was trivial. The targets were counted after plating, the value of the targets' background fluorescence subtracted, a correction of a "standard blank" applied (see Section 3 below) and then the number of counts remaining was divided by the number of micrograms of Yb in the solution that was used to plate the individual targets. The average value of these six

targets became the "standard" number of counts per microgram Yb used in the analysis of vaporization-plated targets.

#### Target Analysis

The method of analysis for vaporization-plated targets is essentially the same as that for the calibration targets given above. That is, it is necessary to find the background fluorescence of the targets before the vaporization run. Once plated the targets are counted again. The difference between the background fluorescence and the plated target's fluorescence should give the fluorescence (as indicated by number of counts the detector sees). Due to instrumental and environmental conditions which change from day to day it is necessary to apply a correction factor in the analysis. One obtains the correction factor by counting one particular target blank (unplated) both before the vaporization run (with the other blank targets) and after the experiment is complete when all the other targets are plated. The difference of its background fluorescence is a direct measure of systematic changes so that the standard blank correction can be defined as the value of (standard blank fluorescence counts before vaporization experiment -- standard blank

fluorescence count after vaporization experiment). The so-called "standard blank" correction is then added to the value of the fluorescence of Yb on each individual target to ascertain the correct value of mass of Yb on each target.

#### CHAPTER IV

#### RESULTS AND DATA REDUCTION

# A. Elemental Analysis of $YbBr_2$

Table I shows the results of chemical analyses of  ${\tt YbBr}_2$  prepared for the vaporization experiments.

Table I. Analytical Results

Analysis wt		% Yb		wt % Br	
#		Theoretica	1	Theoretical	
1	50.46	51.99	48.74	48.01	
2	50.48		48.63		
3	51.61		48.90		
Average	50.8 <sub>5</sub> ±0.7 <sub>0</sub>		48.7 <sub>6</sub> ±0.1		
These val	ues correspo	nd to the	formula YbBr <sub>2</sub>	2.0 <sub>0</sub> ±0.1,	

## B. X-Ray Fluorescence Calibration

A plot of counts of  $L_{\alpha 1}$  radiation from Yb versus  $\mu g$  of Yb was linear over the range of 2-12  $\mu g$  ytterbium.

Table II shows the data collected at each concentration and the average number of counts per microgram ytterbium. The average value was used to obtain mass values used in II-7 which subsequently generated the equilibrium vapor pressures listed in Appendix A.

Table II. X-Ray Fluorescence Calibration Results

μg Yb	Counts 1 µg Yb	
2.8	1857.1	
4.7	1599.8	
6.5	2076.2	
8.4	2506.9	
10.3	2516.7	
12.1	1662.8	
Average value = 2037±405		

### C. Thermocouple Calibration

Numerous attempts were made to calibrate the chromelalumel thermocouple against the melting points of aluminum, silver, lead, and copper. In the first three cases either the heating rate could not be slowed enough to observe the melting point transition or the metal interacted (alloyed) with the molyendum crucible. With copper a transition was seen in the temperature versus time plot which corresponded well with the melting point reported for the copper sample by the National Bureau of Standards (1083.3°C). separate meltings were made with deviations of 0.8, 0.6, and 0.3°C. It is believed that some interaction was beginning between the sample and the crucible and that the first value of 0.8°C represented the true deviation. no other metals were readily available a quick qualitative check calibration was made against the ice point and boiling point of water. The thermocouple showed readings which averaged approximately +1.5° and it was judged that no significant error in temperature measurement would occur if readings were taken directly with no correction applied for the vaporization experiments (±1.5° presents only a 0.1% error in temperature at 1360°).

## D. Vaporization Experiments

Six independent vaporization experiments were carried out with two different Knudsen cells (different size orifice). All of the experiments were effected in a molyendum cell except number 6 in which the sample was placed in a graphite holder with the usual molyendum end caps. Table III lists the orifice size and temperature range of each vaporization experiment.

		1

Table III. Vaporization Experiments

Experiment #	Area of Orifice (x10 <sup>4</sup> )	Temperature Range
1	$9.37 \pm 0.01 \text{ cm}^2$	1207-1296 к
2	9.37±0.01	1190-1279
3	2.94±0.02	1337-1359
4	3.01±0.01	1300-1478
5	2.78±0.00	1309-1514
6	2.90±0.02	1407-1482

The results of the vaporization experiments are graphically represented in Figure 6.

#### E. Mode of Vaporization

Inspection of the X-ray powder diffraction patterns for the sample of starting material and product remaining in the cell after vaporization cataloged in Appendix E shows that YbBr<sub>2</sub> was the predominate chemical substance both before and after the experiments. Indeed the pattern showing spurious lines is that of the starting material. These lines seem to have disappeared in the product left after vaporization; indicating that, whatever their cause, the substance has been destroyed by the heating process. By looking at known powder patterns of Yb(BrO<sub>3</sub>)<sub>3</sub> · 9H<sub>2</sub>O,

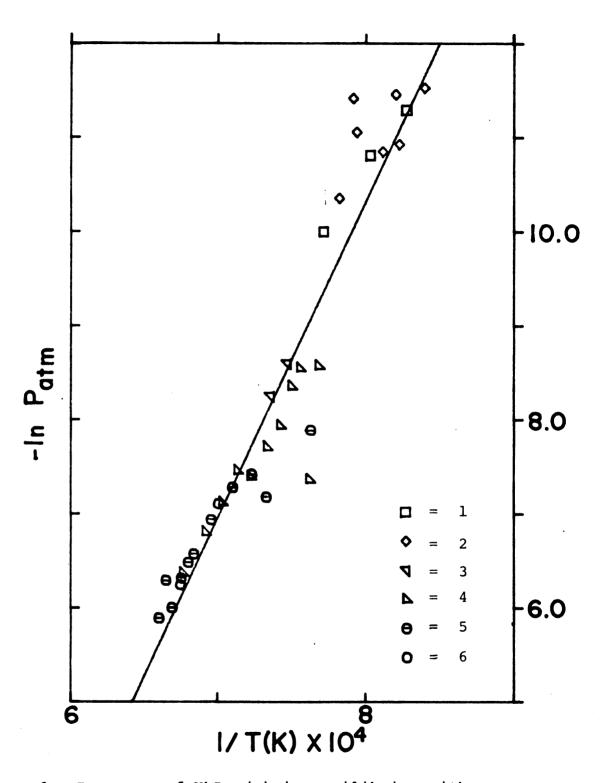


Figure 6. Pressure of YbBr<sub>2</sub>(g) in equilibrium with YbBr<sub>2</sub>(s).

Yb<sub>2</sub>O<sub>3</sub>, Yb<sub>3</sub>O<sub>4</sub>Br, and YbOBr<sup>32</sup> it was evident that the extra lines in the starting material could not be ascribed to any of these compounds. Comparison of the process which yield the starting material revealed that the only other possible contaminates would be ZnBr2 or YbBr3. Of these the first would be eliminated from a YbBr, melt very early in the vaporization process because of its appreciable vapor pressure (just such a process of heating under high vacuum was used to supposedly preferentially remove ZnBr, from the crude YbBr<sub>2</sub> in the reaction sequence used to produce  $YbBr_2$ ). Most probably the contaminate is  $YbBr_3$  which "bumped" over during the distillation process used to purify crude YbBr, in production of the starting material. As a measure of its possible quantative contamination of the starting material one need only look to the elemental analysis of YbBr, described in IV-A. Here it is shown that the starting material is slightly bromine rich (compare molecular formula  $YbBr_{2.0_8}$ ) but a possibility exists for contamination by slight amounts of YbBr3. Indications are the  $YbBr_3$  decomposes under heat 2 thus giving a possible explanation to the pattern of  $YbBr_2$  after the vaporization process has been accomplished.

In view of the powder patterns, elemental analysis, and lack of a trend in third law measurements (see IV-K) the mode of vaporization can be ascribed to a congruent process whereby  $YbBr_2(\ell)$  vaporizes to  $YbBr_2(g)$ .

# F. Vapor Pressure Equation -- $\Delta H^O$ and $\Delta S^O$ at Mean Temperature of Vaporization Experiments

When data of vapor pressure and reciprocal temperature are treated by least squares regression equation (IV-1) results:

$$\ln P_{\text{YbBr}_2} = -(3.37 \pm 0.18) \times 10^4/\text{T} + 16.6_1 \pm 1.3_2$$
(IV-1)

in the temperature range 1190 < T < 1514 K. For comparative purposes the vapor pressure equation for the experiments with the larger orifice (numbers 1 and 2) is presented as:

$$ln P_{YbBr_2} = -(1.7_0 \pm 0.5_4) \times 10^4/T + 2.8 \pm 4.4$$
 (IV-la)  
 $1190 < T < 1295 K$ 

and the vapor pressure equation obtained by the remainder of the experiments with the smaller orifice is:

$$\ln P_{\text{YbBr}_2} = -(2.2_4 \pm 0.1_9 \times 10^4)/T + 8._7 \pm 1._3 \text{ (IV-lb)}$$

$$1300 < T < 1514 \text{ K}$$

however, it is equation IV-1 which is used in all subsequent data treatments since, indeed, the third law data treatment (Section IV-K) shows that the values between the

two sets of data represented by IV-la and IV-lb are coherent with no observable temperature trend. The mean temperature is 1360 K so that through II-l3 and II-l4  $\Delta H_{1360}^{O}$  and  $\Delta S_{1360}^{O}$  are calculated as

$$\Delta H_{1360}^{O} = 66._9 \pm 3._5 \text{ kcal/mole}$$

$$\Delta S_{1360}^{0} = 33._{0} \pm 2._{6} \text{ eu}$$

#### G. Estimation of Thermodynamic Values

The absence of experimentally determined enthalpies and entropies makes it necessary to resort to estimates of these values for the data reduction process.

If  $(H_T^O - H_{298}^O)$  and  $(S_T^O - S_{298}^O)$  functions are available for the various phases or can be estimated reasonably then the relationships:

$$(H_T^O - H_{298}^O) = \int_{298}^T Cp \ dT$$
 (IV-2)

and

$$(S_T^O - S_{298}^O) = \int_{298}^T Cp/T dT$$
 (IV-3)

allow substitution into II-15 and II-16 to give:

$$\Delta H_{298}^{O} = \Delta H_{T}^{O} + \left[\sum_{i} v_{i} \left(H_{T}^{O} - H_{298}^{O}\right)_{i} - \sum_{j} v_{j} \left(H_{T}^{O} - H_{298}^{O}\right)_{j}\right]$$
(IV-4)

and

$$\Delta s_{298}^{o} = \Delta s_{T}^{o} + \left[\sum_{i} v_{i} \left(s_{T}^{o} - s_{298}^{o}\right) - \sum_{j} v_{j} \left(s_{T}^{o} - s_{298}^{o}\right)_{j}\right]$$
(IV-5)

where  $\nu_i$  refers to coefficients on products i, and  $\nu_j$  refers to reactants j.

When necessary entropies can be found in a similar manner by:

$$\Delta s_{298}^{o} = \sum_{i} v_{i} s_{298_{i}}^{o} - \sum_{j} v_{j} s_{298_{j}}^{o}$$
 (IV-6)

In vaporization reactions only one reactant exists so that the summation on j is dropped:

$$S_{298_{\dot{1}}}^{\circ} = 1/v_{\dot{1}} (\sum_{i} v_{i} S_{298_{\dot{1}}}^{\circ} - \Delta S_{298}^{\circ})$$
 (IV-7)

Absolute entropy can also be calculated in a purely estimative manner by the following relationship due to Latimer <sup>33</sup> as reinvestigated by Gronwold and Westrum. <sup>34</sup> It is:

$$S_{298}^{O} = \sum_{i} v_{i} S_{i}^{O} + \sum_{i} v_{i} M_{i}$$
 (IV-8)

where  $S_{i}^{O}$ 's are the lattice contributions and Mi's are the magnetic contributions to entropy.

Finally thermodynamic values for gases can be estimated by choice of a proper model system. Haschke  $^{35}$  reviews the method as applied to rare earth systems, where basically it is established that thermodynamic functions for gases rely mostly on molecular symmetry and not mass effects. Following his arguments for EuBr<sub>2</sub>, YbBr<sub>2</sub> is assumed to have D<sub> $\infty$ h</sub> symmetry, and, therefore, values of HgBr<sub>2</sub>(g) are chosen for the model system.

Estimated values for gases are those given in the JANAF<sup>36</sup> table, while  $(H_{\rm T}^{\rm O}-H_{298}^{\rm O})$  and  $(S_{\rm T}^{\rm O}-S_{298}^{\rm O})$  for solids were found by graphical interpolation of estimates for YbBr<sub>2</sub> found in Brewer et al.<sup>2</sup> and Bulletin 605 of the National Bureau of Mines<sup>37</sup> (see Appendix B).

#### H. Second Law Data Reduction to 298 K

Data from the second law treatment is reduced to from the mean temperature of 1360 K to a reference temperature of 298 K through the use of values compiled in Appendix B and similar values for HgBr<sub>2</sub>(g) by application of IV-4 and IV-5:

$$\Delta H_{298}^{O} = 66.89 + 28.64 - 15.70 = 79.8 \pm 4.4 \text{ kcal/mole}$$

$$\Delta S_{298}^{0} = 33.01 + 36.9 - 22.36 = 47.5 \pm 3.9 eu$$

where errors were assumed to be 20% in the values in Appendix B.

#### I. Value of Absolute Entropy

The absolute entropy of YbBr<sub>2</sub> was found from IV-8 by the use of Latimer's<sup>33</sup> value for Br<sup>-</sup> lattice contribution (10.9 eu) and Westrum's<sup>38</sup> value for Yb(II) lattice contribution (13.1 eu). No magnetic contribution was assumed since Yb(II) has a fully filled 4f shell.

The value of absolute entropy thus obtained was,  $S_{298}^{O} = 13.1 + 2(10.9) = 34.9 \text{ eu.}$ 

# J. Estimation of fef and $\Delta$ fef for YbBr $_2$

So that the tabulated data computed in Appendix B can be used for data reductions by the third law technique it is necessary to rewrite II-19 as:

$$fef = (H_{T}^{O} - H_{298}^{O})T - (S_{T}^{O} - S_{298}^{O}) - S_{298}^{O}$$
 (IV-9)

Use of absolute entropy and values of  $(H_{\rm T}^{\rm O}-H_{298}^{\rm O})$  and  $(S_{\rm T}^{\rm O}-S_{298}^{\rm O})$  from Appendix B generates  $-(G_{\rm T}^{\rm O}-H_{298}^{\rm O})/{\rm T}$  data since it is just equal to fef (see II-18). The use of

II-20, through combination of  $(G_{\rm T}^{\rm O}-H_{298}^{\rm O})/{\rm T}$  data of YbBr<sub>2</sub>(s) in Appendix B and similar data for the vapor assumed equal to that of HgBr<sub>2</sub>(g)<sup>36</sup> allows calculation of estimated  $\Delta$ fef for YbBr<sub>2</sub> vaporizations (Appendix C).

## K. Data Reduction by the Third Law Procedure

Values of Afef from Appendix C were fitted by least squares regression to a parabola of the form:

$$\Delta fef = aT^2 + bT + c \qquad (IV-10)$$

to yield; a =  $-1.2 \times 10^{-5}$ , b =  $3.78 \times 10^{-2}$ , c = -66.12. An excellent fit of data was obtained with the largest deviation for any data point being 0.51 eu. Thus obtained, a, b, c along with equation IV-10 when coupled with  $\ln P_T$  and temperature values in equation II-22 yield  $\Delta H_{298}^O$  values according to the third law treatment. The resulting average value,  $\langle H_{298}^O \rangle = 72._3 \pm 1._5$  kcal/mole with no apparent trend with temperature over the 324 degree range of the vaporization.

## L. Other Thermodynamic Parameters for $YbBr_2$

#### 1. From Literature Estimates and Measured Data

Knowledge of  $\Delta H_{298,V}^{O}$  from the  $\ell n$  P<sub>T</sub>-inverse temperature plot along with literature values of enthalpies of formation of Yb(g)<sup>39</sup> and Br(g)<sup>39</sup> combined with the dissociation energy of YbBr<sub>2</sub><sup>40</sup> permits calculation of the estimated enthalpy of formation of YbBr<sub>2</sub>(g):

$$\Delta H_{f298}^{O} = \sum_{i} v_{i} \Delta H_{298_{i}}^{O} - \sum_{j} v_{j} \Delta H_{f298_{j}}^{O}$$
 (IV-11)

where, again,  $v_i$  refers to coefficients of products and  $v_j$  to those of the reactants:  $\Delta H_{298}^O$  YbBr<sub>2</sub>(g) = -90.2 ± 0.2 kcal/mole.

For the special case of vaporization IV-11 reduced to:

$$\Delta H_{f298}^{O} = 1/\nu_{j} (\sum_{i} \nu_{i} \Delta H_{f298_{i}}^{O} - \Delta H_{298}^{O})$$
 (IV-12)

When i refers to the gas phase and  $\Delta H_{298}^{O}$  equals the second law enthalpy of vaporization IV-12 yields  $\Delta H_{f298}^{O} \text{ YbBr}_{2}(s) = -170 \cdot_{0} \pm 4 \cdot_{4} \text{ kcal/mole.}$ 

The second law absolute entropy of vaporization was determined from the entropy of vaporization and entropy of gaseous YbBr<sub>2</sub> (found for the model HgBr<sub>2</sub> gas<sup>36</sup>) through IV-7. It is,  $S_{298}^{\circ}$  YbBr<sub>2</sub>(s) = 29.<sub>0</sub> ± 3.<sub>9</sub> eu.

The value of the absolute entropy combined with the entropies of Yb(s) and  $Br_2(\ell)$  (see Appendix D) in

$$\Delta S_{f298_{i}}^{O} = S_{298_{i}}^{O} - \sum_{j} v_{j} S_{298_{j}}^{O}$$
 (IV-13)

where  $\nu$  denotes coefficients on reactants j in formation of compound i, yields  $\Delta S_{f298}^{O}$  YbBr<sub>2</sub>(s) = -21.<sub>7</sub> ± 3.<sub>9</sub> eu.

Relationship II-11 allows calculation of  $\Delta G_{f298}^{O}$  YbBr<sub>2</sub>(s) from  $\Delta H_{f298}^{O}$  YbBr<sub>2</sub>(s) and  $\Delta S_{f298}^{O}$  YbBr<sub>2</sub>(s) as;  $\Delta G_{f298}^{O}$  YbBr<sub>2</sub>(s) = -163.<sub>5</sub> ± 4.<sub>6</sub> kcal/mole. Further  $\Delta S_{f298}^{O}$  YbBr<sub>2</sub>(g) can be found from  $\Delta S_{f298}^{O}$  YbBr<sub>2</sub>(s) and the entropy of vaporization as:  $\Delta S_{f298}^{O}$  YbBr<sub>2</sub>(g) = 25.<sub>8</sub> ± 3.<sub>9</sub> eu. Combination of the entropy of formation of YbBr<sub>2</sub> gas with the estimated enthalpy of formation gives an approximate value of  $\Delta G_{f298}^{O}$  YbBr<sub>2</sub>(g) = -(97.<sub>9</sub> ± 1.<sub>2</sub>) kcal/mole.

## 2. From Extrapolation of the Vapor Pressure Equation

The normal boiling point of YbBr $_2(\ell)$  was found by extrapolation of IV-1 to one atmosphere. It has the value:  $T_b = (2.0_3 \pm 0.1_1) \times 10^3$  K where error is associated with enthalpy only. The data of  $(H_T^O - H_{298}^O)$  for YbBr $_2(\ell)$  were determined in the 1200-3000 K range by choosing a PbBr $_2$  model system. Graphical interpolation yields  $(H_{2026}^O - H_{298}^O) = 46.85$  kcal/mole for the liquid and again by resorting to HgBr $_2$  values for gases,  $(H_{2026}^O - H_{298}^O) = 46.85$ 

25.62 kcal/mole for YbBr<sub>2</sub> gas. From IV-4  $\Delta H_V^O$  is calculated as 58.<sub>6</sub> ± 4.<sub>4</sub> kcal/mole and, since  $\Delta S_V^O = H_V^O/Tb$ ,  $\Delta S_V^O = 28.9_2 \pm 2._2$  eu.

#### M. Note on Errors

The problem of combination of errors in thermodynamic cycles was handled in the preceding sections by a treatment due to Feller.  $^{41}$  In essence if Xi(1  $\leq$  i  $\leq$  n) is a set of variables with associated standard deviations  $\sigma$ i, the deviation in their sum, y, is  $\sigma$ y according to:

$$\sigma y = (\sum_{i=1}^{n} \sigma_i^2)^{1/2}$$
 (IV-14)

Since data used in thermochemical cycles are visually independent IV-14 should be a reasonable estimator of error.

#### CHAPTER V

#### DISCUSSION AND SUGGESTIONS

# A. Preparation of YbBr<sub>2</sub>

The method followed for the preparation of YbBr $_2$  outlined in III-A provided an adequate source of starting material although one final step was added to the process as outlined by DeKock and Radtke,  $^{30}$  namely the distillation of the reduced product. It was found that appreciable quantities of the oxide formed if the system leaked air or if the inert gas used during the high temperature reduction contained any  $^{0}$ 2 or  $^{1}$ 20. The oxide formed as a crust upon the dark green melt. By distillation the separation of the reduced [Yb(II)] bromide could be effected from the oxide.

#### B. X-Ray Fluorescence Procedures

The X-ray fluorescence technique theoretically provides a rapid and direct microanalytical determination.

However, when operated in a non-vacuum mode the actual number of counts the instrument records is dependent on

air pressure. The blank counting procedure should correct for the problem, that is, it should eliminate those systematic errors due to day to day variability of the instrument, but in fact the count rate seemed to vary rapidly -even during a 20 minute analysis. Surely such changes cannot be ascribed to air pressure but other variables not so easily accounted for come into play. It was found that during a long analysis (of the order of 45 minutes) the number of counts steadily decayed. Although initially the standard blank was counted often during the analysis runs it was decided to count it before and after each tar-Still count changes were suspect even during the two minutes needed to analyse each target. In addition the necessity of keeping the counts low (required by remaining in the microgram region) caused any error in counting to be proportionally higher than would the same deviation cause if the counts due to plated material were high. is my belief that whatever instrumental cause is responsible for the count variation is the weak point of the entire experimental procedure. Mechanical means are at hand to insure that assumptions necessary to the techniques are not violated, but no such mechanism exists to remove the heretofore inexplicable drop in count rate. At best one must count only three or four targets then allow the unit to "rest" for at least one hour before continuing the analysis. Even so, moment to moment changes cannot be

accounted for and the most one can hope for is that the variance from the true count not be more than an acceptable percentage.

#### C. Target Collection Knudsen Effusion Technique

The highest vapor pressure attained during the vaporization experiments was  $2.5 \times 10^{-3}$  atm. The value is of the order of magnitude of the upper pressure limit beyond which molecular flow cannot be maintained but not above the pressure limit given by Mayer. 28 (See Section II-E.5.) expected no discontinuity was displayed for values obtained by varying the area of the orifice, thus creditability is given to the assumption that Knudsen conditions of equilibrium were maintained in the effusion cell. However satisfying the second law and third law data seem, one problem remained throughout the experiments. inspection of the interior of the effusion cells after the vaporization it was noticed that at least some material had blocked the orifice of the cell. Such behavior is ascribed to a temperature gradient across the cell by Haschke, 35 a condition thought to be corrected by the placement of the cell assembly in a symmetric oven. Indeed, optical pyrometer measurements indicated that no temperature gradient existed across the oven, top to bottom but of course, no such measurement could be made on the

cell itself. Lack of a trend in the enthalpies derived by the third law data treatment suggests that even if a temperature gradient did exist it had little effect upon vapor pressure.

An alternative to a temperature gradient could be selective condensation of the vapor on the lid of the cell interior during the cooling process. Direct evidence of such a situation is not at hand but it seems reasonable since a possible mechanism exists whereby the top and bottom of the cell might lose energy more rapidly than the sides of the cell. First, the walls of the cell are at least twice as thick as the lid since the sample container consists of an insert and end caps; the walls are composed of one layer of insert and one of end cap while the lid is formed only from the end cap. If the extra mass of metal in the sides retains its heat longer than the smaller mass of the lid then selective condensation might occur. Secondly the oven which is about the cell has only two openings, one at the top directly above the orifice of the cell and one coaxially placed at the bottom of the oven. Since the system cools in a high vacuum the only possible way for the cell and oven assembly to release energy is through infrared radiation. The oven essentially stops direct radiation of the cell and a back radiation system might be established -- except at the lid and bottom of the

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cell where it can lose radiation directly to the external environment. Should either condition lower the temperature of the lid preferentially to the walls then selective condensation could occur.

## D. The Use of Thermodynamic Approximation

## 1. Absolute Entropy Approximation

The value obtained by equation IV-8 for the estimated absolute entropy (34.9 eu) falls outside the range of the calculated absolute entropy (29.0  $\pm$  3.9 eu). The difference could be ascribable to the measured  $\Delta S$  at the median temperature if one can assume the  $HgBr_2$  model system represents  $YbBr_2$ (g) accurately since it is a simple combination of these two entropy values in IV-7 which yields the absolute entropy. Indeed, it was because of the probability of variation in these two absolute entropy values that  $\Delta S_{298}^{O}$ , the calculated (IV-8) value was used in estimation of fef's for the third law determinations.

# 2. YbBr<sub>2</sub> Data

Most of the determined values are consistent with available estimated values. The equilibrium vapor pressure is within the limits put forth by Brewer.<sup>2</sup> Estimation of

the normal boiling point results in a value (2026 K) in good agreement with Brewer's estimate. However the value of  $\Delta S_{\mathbf{v}}^{\mathsf{O}}$  of 28.9 eu is outside the range of Trouton's rule. However, Gschneidner 42 points out that liquids boiling much higher than room temperature tend to have higher entropy of vaporization values and recommends a value of 25.5 eu for such substances. Even so, a high value such as 28.9 eu could only come from a value of  $\Delta H_{\mathbf{v}}^{\mathbf{O}}$  which is too large. Indeed, a scan of the JANAF 36 tables shows a definite trend toward increasing  $(H_{T} - H_{298}^{O})$  values with increasing molecular weight for liquid compounds of metal dibromides. The PbBr, model system was chosen for YbBr,  $(\ell)$ to minimize  $\Delta H_{v}$  to bring it as closely in line with the estimated value as possible. Lack of proper model systems in the rare earth series precludes the strict application of an estimative system.

The  $\Delta H_{298}^{O}$  second law value of  $79._{8} \pm 4._{4}$  kcal/mole is some 8 kcal/mole above the value estimated by Feber  $^{40}$  (71 kcal/mole); the average third law value of  $72._{3} \pm 1._{5}$  kcal/mole is in much closer agreement. The  $\Delta H_{f298}^{O}$  YbBr<sub>2</sub>(s) =  $-170._{0} \pm 4._{4}$  kcal/mole is also somewhat lower than the estimated value of -161 kcal/mole  $^{40}$  again probably due to the use of the  $\Delta H_{298}^{O}$  (second law) value which is higher in magnitude than it should be.

The overall magnitude of error in second law values can be shown by examination of IV-1, IV-la, and IV-lb.

Here one sees that the error in the definition of a least squares line can greatly affect the slope (and hence the enthalpy) and the intercept (thereby the entropy). Indeed even if the slope is known to within a small error probability the variance in intercept could be appreciable since the slope is of so great a magnitude. As stated before, the accuracy of the system rests on the consistency shown by the third law treatment without which one would be hard pressed to find reason to join data which forms the two equations (IV-la and IV-lb) into an overall vapor pressure equation (IV-l).

Ultimately the correctness of any data set can only be ascertained by comparison to the trends established by other researchers when no mechanism exists to ascertain in an absolute fashion the necessary relationships.

To address the question of molecular geometry, DeKock and Wesley 43 have linked the degree of non-linearity in the rare earth halides to the s-d orbital separation through actual measurement of infrared spectra of some divalent rare earth halides. On passing from the difluoride to the dichloride the bond angle opened for samarium and europium by some 20°, but for some unexplained reason ytterbium remains invarient with increasing molecular weight (greater s-d orbital separation) of the attached halide. Nonetheless they point out the strong correlation between the

alkaline earth halides and those of the rare earths and it was upon that basis that a linear structure was chosen for YbBr<sub>2</sub>.

Thermodynamic vaporizations have been carried out on both  ${\rm YbCl_2}^{17}$  and  ${\rm YbF_2}^{44}$ . The latter case proved to be incongruent but many of the vaporization parameters were nonetheless described or estimated. Table IV presents values from these references and those found by Hariharan for the analogous Eu(II) compounds.

Table IV. Data Summary for Vaporization Thermodynamics of Eu(II) and Yb(II) Halides

	MF <sub>2</sub>	MCl <sub>2</sub>	MBr <sub>2</sub>	MI <sub>2</sub>
ΔH <sub>298</sub> kcal/mole Eu(II)	100. <sub>8</sub> ±2. <sub>5</sub>	84·9 <sup>±1</sup> ·1	79. <sub>8</sub> ±1. <sub>5</sub>	75. <sub>4</sub> ±1. <sub>1</sub>
$\Delta H_{298}$ kcal/mole Yb(II)	109. <sub>0</sub> ±8. <sub>5</sub>	84.7 <sup>±1</sup> .1	79. <sub>8</sub> ±4. <sub>4</sub>	
ΔS <sub>298</sub> eu Eu(II)	48·9 <sup>±2</sup> ·5	48.3 <sup>±1.</sup> 8	48·0 <sup>±2</sup> ·0	48·1 <sup>±1</sup> ·9
ΔS <sub>298</sub> eu Yb(II)		48·0 <sup>±1</sup> ·1		
ΔH <sup>O</sup> <sub>V</sub> kcal/mole Eu(II)	79·3 <sup>±2</sup> ·0	58.9±1.8	55.3 <sup>±2</sup> .2	53. <sub>5</sub> ±1. <sub>6</sub>
ΔH <sup>O</sup> <sub>V</sub> kcal/mole Yb(II)		61.7 <sup>±1.</sup> 0	58.6 <sup>±4</sup> .4	
$\Delta S_{\mathbf{v}}^{\mathbf{O}}$ eu Eu(II)	31. <sub>7</sub> ±1. <sub>0</sub>	25.2 <sup>±0</sup> .7	<sup>25</sup> ·1 <sup>±1</sup> ·0	26.1 <sup>±0</sup> .7
ΔS <sup>O</sup> <sub>V</sub> eu Yb(II)		26.7 <sup>±0</sup> .7		

In view of these data it is evident that the YbBr<sub>2</sub> system presents values which are in line with those derived for related systems. Even though the amount of scatter in the data points and the resultant error would indicate that the overall value should be somewhat suspect, the system proved to be well-behaved in that consistent values were obtained which allowed the description of the vaporization parameters of YbBr<sub>2</sub>.

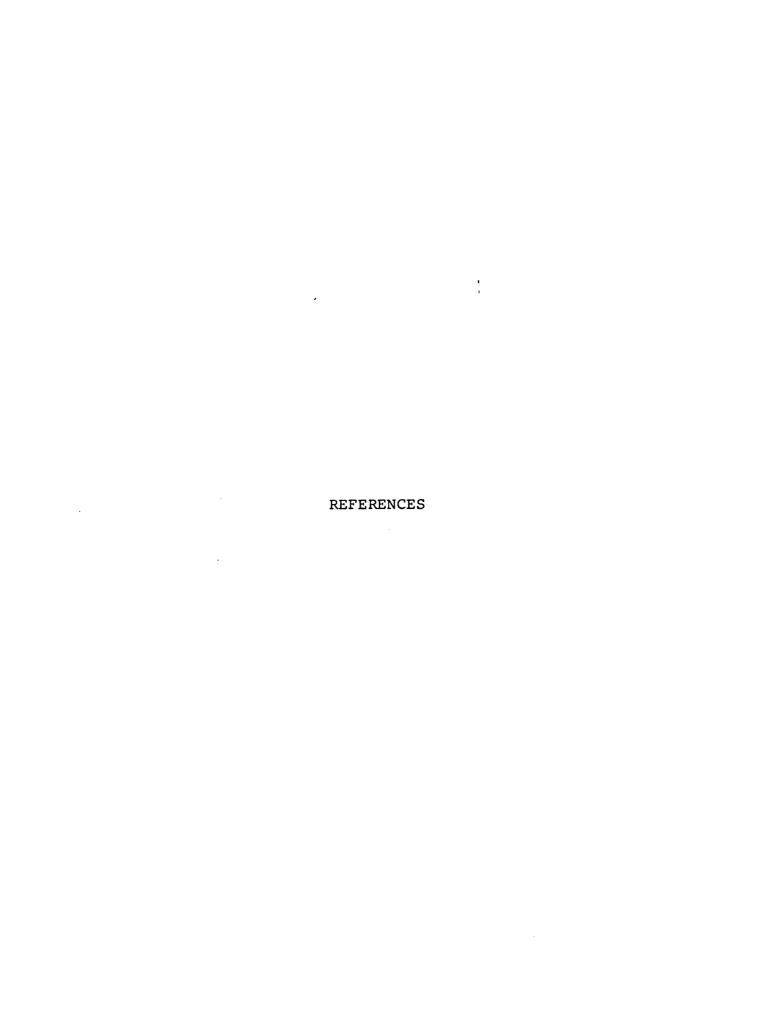
### E. Suggestions for Future Research

Examination of Figure 5 shows considerable variance in the individual data points. In addition those with most average variance are at the lower temperatures. These points were taken early in the experiments before partial remedies were found for the counting technique problem with the X-ray fluorescence unit and before it was determined just how long to expose targets to obtain plated quantities in the calibration region. While there are enough points to establish that no trend in measurement exists I felt it invalid to eliminate any of them through statistical means since such a procedure would have negated the overall importance of the initial experiments which were intended to have equal weight given to them as latter experiments. At present it is unknown whether or not the linearity of the data at higher temperatures is due to a variable in the

vaporization process or elimination of some random errors due to improvement in the skills of the investigator. Obviously what is needed is additional experiments at the lower temperatures so that more data can be obtained for that region. Also the gap between the lower and higher temperatures could be filled in since it is the region where one would expect a cell with an orifice of intermediate area between those used to supply data. Regardless of the amount of data obtained the enthalpies and entropies at a reference temperature will only be as reliable as the system used for reduction to that temperature from the operating temperature of the experiments. Since the choice of the HgBr, model system for YbBr, (g) was dependent on the necessity for  $\mathbf{D}_{\infty\mathbf{h}}$  symmetry should the  $\mathbf{YbBr_2}$  molecule prove to be non-linear then revaluation of the system with a new model system would be necessary.

Finally it should be pointed out that at least one other system exists for estimation of the thermodynamic properties of materials. Through theories developed by statistical thermodynamics, known rotational, vibrational and electronic states of a molecule allow calculation of its heat capacity,  $(H_T - H_{298}^O)$  and  $S_T^O$  functions. Although more pleasing in essence, it too is dependent on assumptive techniques when the necessary preliminary data from measured spectra are not known. Hariharan has applied the method to Haschke's  $^{17,35}$  data for EuBr<sub>2</sub> and obtains

somewhat differing values. His treatment, while more eloquent in nature than extrapolation of estimated thermodynamic values, is only as reliable as his estimates from the extrapolation of data for the alkaline earth bromides. Of course should the electronic, vibrational and rotational spectra of YbBr<sub>2</sub> become available then the system would yield absolute data reduction from median temperatures to any reference temperature.



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APPENDICES

Appendix A: Collected Vaporization Data and Third Law Enthalpies for YbBr<sub>2</sub>

T(K)	μg YbBr <sub>2</sub>	−ln P <sub>T</sub>	Orifice Area (x10 <sup>4</sup> )	Collection Times	<sup>ΔH</sup> 298
1295	19.30	9.999 atm	9.37 cm <sup>2</sup>	45.16 min	73.997 kcal/mole
1208	3.67	11.293		30.24	42.949
1245	5.84	10.812		30.18	73.607
1259	4.52	11.061		30.15	74.921
1278	9.09	10.357		30.22	74.085
1231	5.66	10.852		30.30	73.017
1215	5.24	10.931		30.13	72.423
1190	2.91	11.530		30.15	72.616
1219	3.09	11.460		30.21	73.901
1262	3.26	11.418		31.12	75.966
1338	15.93	8.600	2.94	30.13	72.377
1359	22.37	8.256		30.22	72.428
1311	19.55	7.370	3.01	10.455	67.925
1300	8.55	8.576		15.21	70.563
1322	5.84	8.550		10.21	71.504
1346	10.66	7.941		10.22	70.986
1332	7.02	8.359		10.17	71.460
1363	13.24	7.712		10.16	71.140
1384	14.41	7.400		8.16	71.237
1401	8.45	7.464		5.13	72.186

Appendix A (Cont.)

T(K)	μ <b>g YbBr</b> 2	−ln P <sub>T</sub>	Orifice Area (x10 <sup>4</sup> )	Collection Times	ΔH <sup>O</sup> 298
1421	11.91	7.123 atm	3.01 cm <sup>2</sup>	5.18 min	72.141 kcal/mole
1445	9.84	6.812		3.16	72.348
1477	14.89	6.374		3.12	72.535
1310	10.42	7.891	2.78	10.16	69.238
1364	16.67	7.181		8.15	69.746
1382	9.85	7.424		6.18	71.213
1407	9.32	7.284		5.13	71.957
1437	8.04	6.943		3.18	<b>72.3</b> 59
1462	7.73	6.577		2.14	72.444
1470	8.37	6.486		2.12	72.543
1480	7.52	6.320		1.62	72.513
1494	6.46	6.005		1.02	72.219
1514	5.13	5.894		0.73	72.800
1503	4.34	6.296		0.92	73.498
1407	9.89	7.286	2.90	5.12	71.962
1427	9.25	7.110		4.04	72.378
1481	8.12	6.250		1.53	72.353

Appendix B: Enthalpy, Entropy, Free Energy Functions of  $YbBr_2(s, \ell)$ 

T(K)	(H <sub>T</sub> - H <sub>298</sub> )	$(s_{\rm T}^{\rm o} - s_{298}^{\rm o})$	$-(G_{\rm T}^{\rm O}-H_{\rm 298}^{\rm O})/{\rm T}$
1000	20,000 cal/mole	29.00 eu	43.90 eu
1100	22,400	31.50	46.04
1200	24,800	34.00	48.23
1300	27,200	36.00	49.98
1400	29,600	37.50	51.26
1500	32,000	39.00	52.57
1600	34,400	40.00	53.4

Appendix C: Free Energy Function Changes For the Vaporization of  $YbBr_2(\ell)$ 

T(K)	-Afef
1000	40.061 eu
1100	38.927
1200	37.689
1300	36.839
1400	36.413
1500	35.914
1600	35.855

Appendix D: Thermodynamic Functions

Phase	Thermodynamic Function at 298 K	Value	Reference
Yb(g)	ΔH <sup>O</sup> kcal/gfw	36.35±0.2	39
Br(g)	$\Delta H_{f}^{O}$ kcal/gfw	26.740	36
Yb(g)	ΔG <sup>O</sup> kcal/gfw	28.285	39
Br(g)	ΔG <sup>O</sup> kcal/gfw	19.700	36
Yb(s)	S <sup>O</sup> eu	14.30±0.04	37
$\mathrm{Br}_2(\ell)$	S <sup>O</sup> eu	36.384	36
YbBr <sub>2</sub> (g)	D <sub>o</sub> kcal/gfw	(180)	40

Appendix E: X-Ray Powder Diffraction Patterns

5.17a       40%       5.79       diffuse         4.78       30       4.76       90%       4.79       100%         4.22a,b       40       4.24b       very       xeak       100%       24.7         4.10a       30       3.63       40       3.65       24.7         3.55       20       3.21       50       3.23       23.6         3.12       30       3.21       50       3.07       11.4         2.91       20       2.91       20       2.99       0.3         2.54a,b       40       2.59b       40       2.59       0.3         2.55a,b       60       2.49       30       2.51       26.3         2.45       100       2.45       60       2.47       53.7         2.29       20       2.37       20       2.19       16.1         2.29       20       2.21       10       2.19       16.1         2.29       20       2.21       20       2.19       16.1         2.29       20       2.21       20       2.19       16.1         2.29       20       2.21       20       2.19       16.1	YbBr <sub>2</sub> Sample Before Vap. Exp. d-spacings	Relative Intensity of Reflection	Product Remaining in Cell After Vap. Exp. d-spacings	Relative Intensity of Reflection	Calculated d-spacings for YbBr <sub>2</sub> *	Calc.* Intensity of Reflection	Reflection (h k $\ell$ )
30       4.74b       very very very         30       3.63       40       3.65         20       3.63       40       3.65         30       3.21       50       3.23         30       3.00       10       3.07         40       2.91       20       2.99         40       2.59b       40       2.99         60       2.49       30       2.51         100       2.45       60       2.47         60       2.37       20       2.40         20       2.21       10       2.19         20       2.21       10       2.19		40%	5.79	diffuse			
40     4.24b weak     very weak       30     3.63     40     3.65       20     3.21     50     3.23       30     3.20     10     3.07       20     2.91     20     2.99       40     2.59b     40     2.99       60     2.49     30     2.51       100     2.45     60     2.47       60     2.37     20     2.40       20     2.21     10     2.19       20     2.21     20     2.18		30	4.76	806	4.79	100%	110
30 30 3.63 40 3.65 30 3.21 50 3.23 30 2.91 20 2.91 40 2.59 40 2.59 40 2.59 40 2.59 40 2.59 2.99 40 2.59 40 2.59 40 2.59 40 2.99 2.99 40 40 2.99 40 40 40 40 40 40 40 40 40 40 40 40 40	a	40	4.24 <sup>b</sup>	very weak			
203.63403.65303.21503.23303.00103.07202.91202.99402.59402.99602.49302.511002.45602.47602.37202.40202.21102.19217202.18		30					
30       3.21       50       3.23         30       3.00       10       3.07         20       2.91       2.99         40       2.59       40       2.99         60       2.49       30       2.51         100       2.45       60       2.47         60       2.37       20       2.40         20       2.21       10       2.19         20       2.11       20       2.18		20	3.63	40	3.65	24.7	101
30       3.00       10       3.07       1         20       2.91       2.99       2.99       2.99       2       2.99       2       2.99       2       <		30	3.21	20	3.23	23.6	111
20       2.91       2.99         40       2.59b       40         60       2.49       30       2.51       2         100       2.45       60       2.47       5         60       2.37       20       2.40       2         20       2.21       10       2.19       1         21       20       2.18       2       2		30	3.00	10	3.07	11.4	120
40       2.59b       40         60       2.49       30       2.51       2         100       2.45       60       2.47       5         60       2.37       20       2.40       2         20       2.21       10       2.19       1         217       20       2.18       1		20	2.91	20	2.99	0.3	210
2.49       30       2.51       2         2.45       60       2.47       5         2.37       20       2.40       2         2.21       10       2.19       1         2.17       20       2.18       1	Q	40	2.59 <sup>b</sup>	40			
2.45       60       2.47       5         2.37       20       2.40       2         2.21       10       2.19       1         2.17       20       2.18		09	2.49	30	2.51	26.3	121
2.37     20     2.40     2       2.21     10     2.19     1       2.17     20     2.18		100	2.45	09	2.47	53.7	211
2.21     10     2.19     1       2.17     20     2.18		09	2.37	20	2.40	21.7	220
20 2.18		20	2.21	10	2.19	16.1	005
			2.17	20	2.18	7.6	130

Appendix E (Cont.)

ا سما	i				
Reflection (h k $\ell$ )	310	031	320	222	132
Calc.* Intensity of Reflection	5.3	18.9	1.1	6.2	3.1
Calculated d-spacings for YbBr <sub>2</sub>	2.11	2.04	1.86	1.61	1.54
Relative Intensity of Reflection		20%	30	10	10
Product Remaining in Cell After Vap. Exp. d-spacings		2.03	1.83 <sup>b</sup>	1.60	1.56 <sup>b</sup>
Relative Intensity of Reflection	20%	15			
YbBr <sub>2</sub> Sample Before Vap. Exp. d-spacings	2.08	2.02			

\* Calculated from known X-ray parameters for  ${\tt YbBr}_2$  and  ${\tt CaBr}_2$ .

<sup>&</sup>lt;sup>a</sup>Spurious reflection (see V-E).

 $<sup>^{</sup>m b}_{
m Possibly}$  due to presence of  $^{
m Yb}_2{
m O}_3$  (see Ref. 32).

